

LETTER

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To cite this article: C. Dhanasekhar *et al* 2019 *EPL* **127** 67001

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Structural, magnetic and electric-polarization properties of geometrically frustrated YBaCo_4O_7 and $\text{DyBaCo}_4\text{O}_7$ cobaltites

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received 21 June 2019; accepted in final form 13 September 2019

published online 4 November 2019

PACS 75.85.+t – Magnetoelectric effects, multiferroics

PACS 75.50.Lk – Spin glasses and other random magnets

Abstract – In the RBaCo_4O_7 ($\text{R} = \text{Ca}, \text{Y}$ and rare earth) cobaltite family, only $\text{CaBaCo}_4\text{O}_7$ shows 3D long-range ferrimagnetic and spin-driven electric-polarization properties. In the present study, we have investigated the structural, magnetic and electric polarization properties in YBaCo_4O_7 and $\text{DyBaCo}_4\text{O}_7$ members of this family and the obtained results are compared with the $\text{CaBaCo}_4\text{O}_7$. The compound YBaCo_4O_7 showed a series of magnetic transitions in agreement with the reported literature, additionally, a cluster glass behavior below 5 K is observed. Powder neutron diffraction studies on $\text{DyBaCo}_4\text{O}_7$ cobaltite showed an orthorhombic $Pbn2_1$ symmetry and signature of short-range 120° magnetic correlations of kagome layer and a spin glass behavior below 65 K. Dielectric measurements on both the samples showed strong frequency dispersion at high-temperature and frequency-independent behavior at low temperature without dielectric anomaly. Pyroelectric current measurement has showed a broad peak around 50 K in both the samples; however, a careful analysis relates the peak to thermally stimulated depolarization current. This study signifies that the giant structural distortions and cobalt charge ordering at the kagome layer are the key factors to drive both long-range magnetic ordering and spin-driven electric polarization in this cobaltite family.

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Introduction. – Cobalt oxides consistently showed much attention because of their rich variety of fascinating physical properties such as unconventional superconductivity [1], large thermoelectric power [2], spin state transitions [3], giant magnetoresistance [2,4], charge stripes [5]. In recent years diverse interesting features, *i.e.*, spin liquid phase [6] unidirectional THz absorption and magnetoelectric effects [7,8], and giant magnetic-field-induced polarization [9] were reported in RBaCo_4O_7 ($\text{R} = \text{Ca}, \text{Y}$ and rare earth) cobaltites. The crystal structure of these materials is built up with alternating kagome and triangular layers, where the corners of the triangles are occupied by magnetic cobalt ions and causes the frustration [10,11].

$\text{CaBaCo}_4\text{O}_7$ (CBCO) cobaltite belongs to the noncentrosymmetric orthorhombic structure (space group $Pbn2_1$) and was reported to undergo a 3D ferrimagnetic ordering

below $T_c = 70$ K [10]. The ferrimagnetic ordering arises from the large structural distortion at kagome layers and strong spin exchange interactions between the triangular and kagome layer cobalt atoms along the c -axis. This compound shows a spin-driven electric polarization below 70 K and is found to be pyroelectric in nature (electric polarization cannot switch with the electric field) [9,12–14]. In our previous work, we have demonstrated switching from pyroelectric-ferrimagnetic ground state to ferroelectric-antiferromagnetic state in CBCO by replacing Co^{2+} by Ni^{2+} on the kagome layer [15].

On the other hand, YBaCo_4O_7 (YBCO) also possesses noncentrosymmetric with hexagonal/trigonal structure (space group $P6_3mc/P31c$) [11,16–20]. With lowering temperature, a structural phase transition occurs initially at 310 K from the hexagonal to an orthorhombic phase $Pbn2_1$ and finally to a monoclinic structure $P2_1$ below 100 K [19]. The orthorhombic to monoclinic structural

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transition coincides with the onset of spin lattice coupling in this compound. This relieves partly the geometric frustration and drives it to an antiferromagnetically (AFM) ordered state below 100 K followed by a subsequent spin reorientation around 60 K [19,20]. Further, neutron diffraction studies show a strong spin-spin correlation between the triangular and kagome layer spins along the c -axis, whereas the kagome layer (ab plane) spins show a short-range 120° correlation [16,17,20]. Room temperature X-ray diffraction measurements on the $\text{DyBaCo}_4\text{O}_7$ (DBCO) show that the orthorhombic $Pbn2_1$ phase and magnetization measurements show the Griffiths phase at low temperature [21].

To further understand the spin-driven electric-polarization properties in RBaCo_4O_7 cobaltites, it is essential to study the structural, magnetic- and electric-polarization properties of other members of this cobaltite family and compare their results. With this perspective, the YBaCo_4O_7 and $\text{DyBaCo}_4\text{O}_7$ members of this family were selected and we studied their properties in a systematic way. Although these materials have necessary symmetry conditions [19,21] to have the spin-driven electric polarization, one would expect a different origin of electric polarization as in YBaCo_4O_7 , the Y^{3+} ($S = 0$) ion is nonmagnetic and Dy^{3+} ($S = 5/2$) has rare-earth spin contribution. In $\text{DyBaCo}_4\text{O}_7$ one would also expect $4f$ - $3d$ interactions, which may further influence the magnetic ordering and electric-polarization properties, whereas in the case of YBCO the cobalt sublattice only would expect to cause the spin-driven electric polarization.

Experiment details. – The $\text{DyBaCo}_4\text{O}_{7+\delta}$ (DBCO) and $\text{YBaCo}_4\text{O}_{7+\delta}$ (YBCO) samples were synthesized by the solid-state reaction method following a procedure as reported previously [22,23]. Both powders of YBaCo_4O_7 and $\text{DyBaCo}_4\text{O}_7$ were obtained through a two-step solid-state reaction method starting from stoichiometric high-purity amounts of Y_2O_3 , Dy_2O_3 , BaCO_3 , and Co_3O_4 . The first heat treatment was made at 900°C for 12 h. After regrinding the samples and pressing them into a pellet, the second sintering was done at 1100°C for 24 h. After the final reaction, the pellet was quenched to room temperature in order to stabilize the 114 cobaltite phase [23]. The structural and magnetic properties of this cobaltite family are strongly influenced by the oxygen nonstoichiometry. The oxygen stoichiometry of the as-prepared samples was determined by idiometric titration [24], which shows the nearly stoichiometric ($\delta = 0.04$: for the Y sample and $\delta = 0.07$ for the Dy sample) within the experimental limits. Phase purity of the samples is confirmed by X-ray powder diffraction at room temperature. The neutron diffraction measurements on DBCO samples were carried out on the PD2 powder neutron diffractometer ($\lambda = 1.2443 \text{ \AA}$) at Dhruva reactor, Bhabha Atomic Research Centre, Mumbai, India. Rietveld refinement of the X-ray powder diffraction and neutron diffraction patterns were carried out using the FULLPROF program.

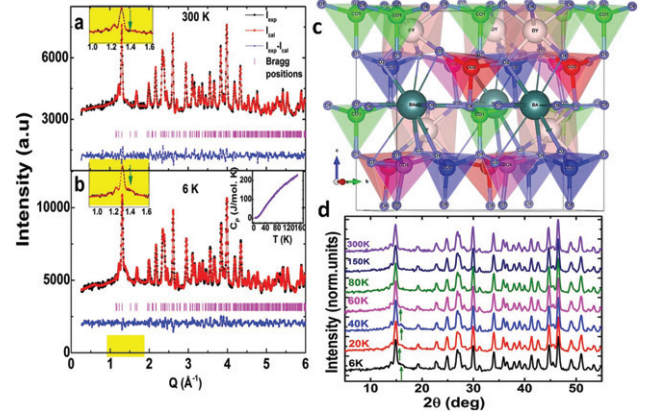


Fig. 1: Room temperature (a) and low-temperature (b) experimental NPD data of the $\text{DyBaCo}_4\text{O}_7$ (black dots) with the simulated curve (red lines). Here, the blue lines show the difference between the experimental and simulated curves and the vertical magenta lines show the corresponding nuclear Bragg positions. The top insets of (a) and (b) show the neutron diffraction patterns in a selected range. The right inset of (b) shows the temperature variation of specific heat. The obtained nuclear structure at 6 K from NPD is shown in (c) and panel (d) shows NPD measured at various temperatures.

DC and AC magnetic measurements were carried in a commercial VSM SQUID magnetometer (Quantum Design, USA). Semiadiabatic heat pulse technique is used to measure the specific heat in the temperature range of 2–150 K. Dielectric- and pyroelectric-current measurements were done in parallel-plate capacitor geometry using Agilent 4291A and Keithley electrometer 6517A, respectively. For dielectric- and pyroelectric-current (I_p) measurements pellets of thickness 0.5 mm and a diameter of 5 mm were used.

Results and discussion. – The phase purity of the YBCO and DBCO samples at room temperature was verified by X-ray diffraction, where YBCO shows a single-phase hexagonal crystal structure with a $P6_3mc$ space group and the DBCO sample shows orthorhombic $Pbn2_1$ symmetry, which are similar to those in the literature [11,16,21]. Neutron powder diffraction measurements on DBCO sample and Rietveld refinement of the NPD patterns recorded at 300 K and 6 K are shown in figs. 1(a) and (b). The experimental data were refined using orthorhombic $Pbn2_1$ symmetry and the obtained nuclear structure of DBCO at 6 K is shown in fig. 1(c). The neutron diffraction pattern on the DBCO sample has shown no increase in the intensity of the fundamental Bragg reflections or superlattice reflections at 6 K indicating the absence of long-range magnetic ordering. However, a broad hump is observed at $Q \approx 1.35 \text{ \AA}^{-1}$ at 6 K and is shown in the inset of fig. 1(b). This broad hump in fact appears below 80 K and is marked by an arrow in fig. 1(d). A similar behavior was observed in the YBCO and other members of this family [11,19,25–27] and their occurrence was assigned to the short-range magnetic correlations.

Table 1: The refined unit cell parameters, lattice volume, orthorhombic distortion (D in %) and reliability factors of $\text{DyBaCo}_4\text{O}_7$ obtained from Rietveld refinement of the neutron diffraction at 300 K and at 6 K. Orthorhombic crystal structure, space group of $Pbn2_1$.

Parameters	300 K	6 K
a (Å)	6.2959(12)	6.2949(12)
b (Å)	10.9200(22)	10.9217(21)
c (Å)	10.2415(8)	10.2158(7)
V (Å ³)	704.120(0.212)	702.365(0.201)
Distortion (D) %	0.153	0.164
R_{Bragg} (%)	4.26	6.08
R_f (%)	2.65	3.93
χ^2	2.35	2.99

In case of YBCO samples, the short-range correlations are followed by long-range magnetic order [16–18], while in the case of DBCO the long-range magnetic order is not established. The lattice parameters, unit cell volume and orthorhombic distortion (D) are given in table 1. At 6 K the b and c lattice parameters are lower than at 300 K, while the a lattice parameter is higher, resulting overall in a small decrease in the unit cell volume.

In the present case, the orthorhombic distortion ($D = (\frac{b}{\sqrt{3}} - a)/a$) is nearly constant from room temperature (0.15%) to 6 K (0.16%) and is much smaller than that of CBCO (1.01% at 300 K and 1.82% at 6 K) [15].

YBCO shows multiple magnetic transitions with varying temperature [11,18]. The $M(T)$ curves show an anomaly at 300 K which can be attributed to the structural phase transition (T_{st}) from the hexagonal phase to the orthorhombic phase. As the temperature is further lowered, magnetization rises below 80 K and shows a maximum at 50 K and then drops down to 10 K, and finally a small increase is found below 10 K. The $M(T)$ behavior above 10 K matches with previous studies, however, the small increase below 10 K is not discussed. The isothermal magnetization ($M(H)$) measured at 5 K is shown in the inset of fig. 2(a). The linear nature of $M(H)$ suggests the AFM nature of YBCO.

The frequency and temperature variation of the in-phase ($\chi'(T)$) component of ac magnetization of YBCO is shown in fig. 2(c). The $\chi'(T)$ curve shows a frequency-independent peak at 80 K, which coincides with the sharp rise of dc magnetization, indicating the antiferromagnetic (AFM) transition. In addition, $\chi'(T)$ shows a broad hump at 50 K and again a sharp peak near 5 K. The peak position in $\chi'(T)$ at 5 K is strongly dependent on the measuring frequency and is shown in the inset of fig. 2(b). Although the dc magnetization studies look similar to the previous studies, $\chi'(T)$ behavior of the YBCO is different from the Valldor *et al.* studies. In Valldor *et al.*'s

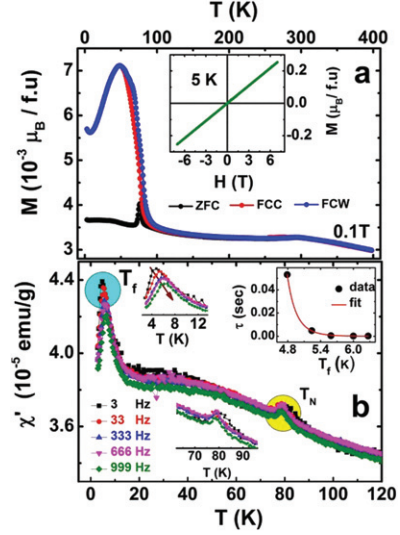


Fig. 2: (a) $M(T)$ curves of YBaCo_4O_7 measured under H_{dc} of 0.1 T and (b) shows the $\chi'(T)$ measured under $H_{ac} \sim 10^{-4}$ T at different frequencies. Inset of (a) illustrates the $M(H)$ curves at 5 K. The right inset of (b) shows the power law fit (to eq. (1)) of low-temperature peak around 5 K.

studies $\chi'(T)$ shows a frequency-dependent peak at 60 K and is assigned to the spin-glass-like behavior. The difference in the $\chi'(T)$ of both studies might be due to the disorder which arises from the changes in the synthesis conditions [16,28].

To understand the nature of this peak we have measured the Mydosh parameter [29]

$$\varphi = \frac{\Delta T_f}{T_f \Delta(\log_{10}(f))} \quad (1)$$

which indicates the relative frequency shift in the peak temperature (T_f) of $\chi'(T)$ per frequency decade and it is found to be ~ 0.094 . In general, the φ value is used to differentiate among various glassy systems; for spin glass $\varphi \sim 0.005$ – 0.01 , for cluster glass $\varphi \sim 0.03$ – 0.06 and in case of superparamagnetism $\varphi > 0.1$ [30]. The obtained φ value in YBCO is higher than that of cluster glass, and this means that the interaction between the clusters is either very weak or completely absent. The frequency-dependent freezing temperature of $\chi'(T)$ could not be fitted to the Arrhenius thermal activation mechanism. The shape of the $M(H)$ at 5 K is also linear, precluding the superparamagnetic (SPM) nature of the YBCO. The relaxation time (τ) was analyzed using the expression for the critical slowing-down power law [29] given by

$$\tau = \tau_0 \left(\frac{T_f}{T_g} - 1 \right)^{-zv}, \quad (2)$$

where τ_0 is the microscopic spin relaxation time, T_g is the glassy freezing temperature and zv denotes the critical exponent. The corresponding fitting is shown in the inset of fig. 2(b), and the obtained values are $\tau_0 = 3.45 \times 10^{-5}$ s,

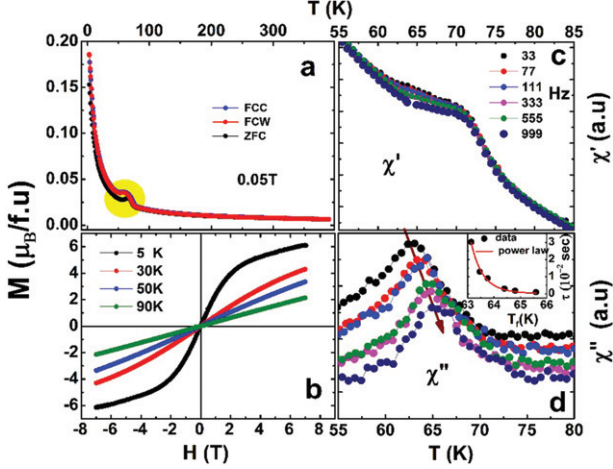


Fig. 3: $M(T)$ curves of $\text{DyBaCo}_4\text{O}_7$ measured under a magnetic field of 0.05 T (a) and $M(H)$ curves recorded at various temperatures (b). Panels (c) and (d) show $\chi'(T)$ and $\chi''(T)$ measured under $H_{ac} \sim 10^{-4}$ T at different frequencies. The inset of inset (d) shows the power law fit (eq. (2)) of maximum $\chi''(T)$ peak temperature.

$T_g = 3.7$ K and $zv = 6$, respectively. The large values of τ_0 and zv indicate that YBCO has the freezing of magnetic clusters rather than the individual atomic spins.

The temperature variation of magnetization of the DBCO sample measured under dc field (H_{dc}) of 0.05 T is shown in fig. 3(a). With decreasing temperature, magnetization shows a peak at 70 K and this behavior is consistent with that in ref. [21], where it was assigned to an AFM ordering of the Co sublattice. However, our NPD study confirms the absence of the long-range magnetic ordering. Further, below 70 K the magnetization increases due to the paramagnetic Dy^{3+} ion contribution.

The $M(H)$ curves measured at various temperatures below and above 70 K are shown in fig. 3(b). At 5 K, $M(H)$ shows a linear and reversible behavior below the critical field (H_C) of 2.2 T and above H_C , $M(H)$ shows a linear behavior up to 7 T without saturation. A similar nonlinear behavior is also found at 30 K, 40 K and 50 K, with decreasing net M and finally a linear $M(H)$ behavior is observed at 90 K which matches with the $M(T)$ behavior. Similar kind of $M(H)$ and H_C values were observed in other cobaltites in this family such as $\text{TbBaCo}_4\text{O}_7$ [31] and $\text{YbBaCo}_4\text{O}_7$ [32] while such a behavior is not observed in YBCO (inset of fig. 2(a)), this indicates that the rare-earth magnetic moment at the “A” site is related to this field-induced change in magnetization.

Further, to understand the magnetization of DBCO samples we have measured the temperature-dependent in-phase ($\chi'(T)$) and out-of-phase ($\chi''(T)$) component of ac susceptibility under $H_{ac} \sim 10^{-4}$ T. The obtained results are shown in figs. 3(c) and (d). The $\chi'(T)$ show a step-like feature in the vicinity of 70 K, which closely matches with $M(T)$. Though it shows a frequency dependence, due to the increasing moment a meaningful analysis could not be

carried. On the other hand, a clear frequency variation is found in the $\chi''(T)$. A finite value of $\chi''(T)$ and the corresponding shift in its peak position to higher temperature with increasing frequency suggest a glassy nature. Experimentally, the freezing temperature (T_f) can be obtained from the maximum of $\chi'(T)$, or from the inflection point (or the maximum) of $\chi''(T)$ [30]. Both methods were extensively used to find the T_f in spin glass (SG) systems [33,34]. Since the maximum of $\chi'(T)$, in our system is difficult to identify, we have used the maximum of $\chi''(T)$ to obtain T_f . The φ has been estimated from the $\chi''(T)$ peak and is found to be ~ 0.019 , which falls within the spin glass range [29].

Further, the relaxation dynamics of the DBCO is examined using the power law given by eq. (2) and the results are shown in the inset of fig. 3(d). From the fitting, the obtained values are $\tau_0 = 1.642 \times 10^{-13}$ s, $T_a = 58.7$ K and $zv = 10.2$. These values are within the range of spin glass materials [29] and, thus, it may be concluded that DBCO is a spin glass. The absence of the long-range magnetic ordering is further supported by the heat capacity measurement with absence of anomaly in the vicinity of 70 K and is shown in the inset of fig. 1(b). The heat capacity behavior matches with the previous studies where no significant anomaly is observed at magnetic ordering temperature [35]. The present ac magnetization and specific-heat measurements complement the absence of long-range magnetic ordering observed from the NPD.

Figures 4(a) and (c) show the real part of dielectric permittivity (ϵ) as a function of temperature for different frequencies in YBCO and DBCO samples. Frequency-independent ϵ was observed at low temperature and a step-like increase with temperature is noticed for both samples. Figures 4(b) and (d) show the temperature-dependent dielectric loss tangent ($\tan \delta$) as a function of frequency. The observed broad peak in $\tan \delta$ shifted with increasing frequency to a higher temperature, indicating a thermally activated relaxation behavior. However, the ϵ shows no peak at the magnetic transition and indicates that there are no permanent electric dipoles or the spin-induced dipoles. The maximum peak position in the $\tan \delta$ is fitted with the thermally activated Arrhenius mechanism (eq. (3)) and is shown in the inset of figs. 4(a) and (d) for the YBCO and DBCO samples,

$$\tau(T) = \tau_0 \exp \left(\frac{E_a}{k_B T} \right), \quad (3)$$

where E_a is the activation energy required for the relaxation process and τ_0 is the pre-exponential factor. The E_a and τ_0 for the DBCO are 0.118(5) eV and 2.91×10^{-11} s and 0.123(1) eV and 5.52×10^{-11} s for YBCO, respectively.

The pyroelectric current (I_p) is recorded while warming the samples, and the obtained results for both samples are shown in figs. 5(a) and (b). In both samples, a broad I_p peak near 45 K is observed, and the maximum peak temperature (T_{max}) depends strongly on the heating rate. The

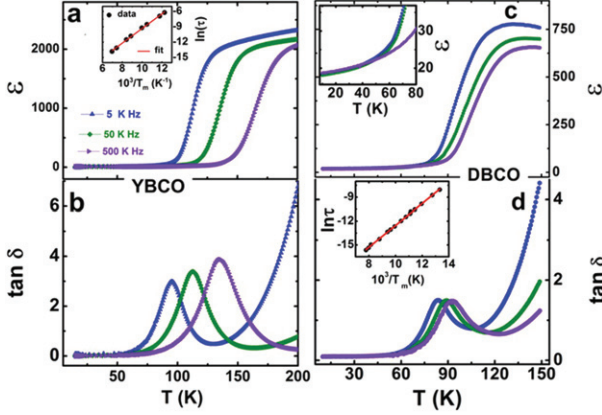


Fig. 4: Panels (a) and (c) show $\epsilon(T)$ curves for the YBaCo₄O₇ and DBCO samples and the corresponding $\tan \delta$ of these samples are shown in (b) and (d). The insets of (a) and (d) show the Arrhenius fit (eq. (3)) for the maximum peak temperature of $\tan \delta$.

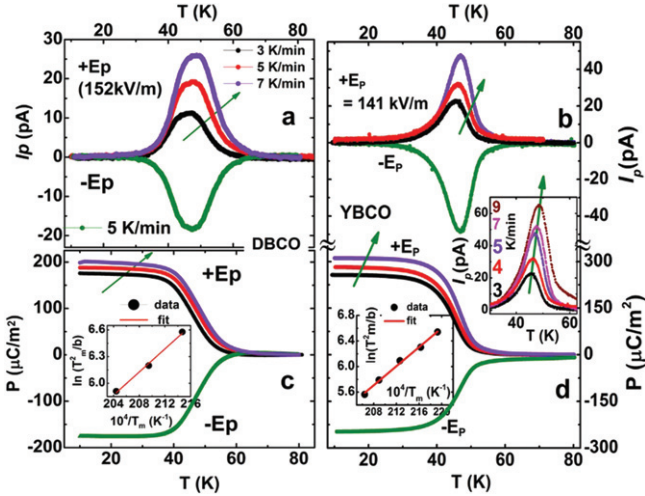


Fig. 5: Panels (a) and (c) show temperature variation of I_p and the corresponding polarization for DyBaCo₄O₇ measured for $E_p = \pm 152 \text{ kV/m}$ under various heating rates. Panels (b) and (d) show I_p vs. temperature and the corresponding polarization for YBaCo₄O₇ measured under $E_p = \pm 141 \text{ kV/m}$. The insets in (c) and (d) show the $\ln(T_m^2/b)$ vs. $1/T_m$ fitting and are described in the main text.

resulting electric polarization (P) for the corresponding I_p is shown in figs. 5(c) and (d) which confirm that the polarizations and their saturation value depend on the heating rate. This behavior is in contrast with CBCO, which shows a sharp λ like the I_p peak in the vicinity of the 65 K and the peak position is found to be independent of the heating rate [15]. The broad I_p peak and its dependence on the heating rate suggest the extrinsic origin of P . This I_p may be due to the thermally stimulated free charge carriers (TSFCC) observed in several spin-driven ferroelectric materials [36,37]. Further, the extrinsic I_p can be understood from the Bucci-Fieschi-Guidi framework [37,38] where the relation between heating rate (b)

and the maximum peak temperature (T_m) of I_p is given by

$$\ln \frac{T_m^2}{b} = \frac{E_a}{k_B T_m} + \ln \frac{\tau_0 E_a}{k_B}, \quad (4)$$

where E_a is the activation energy, τ_0 is the relaxation time and b is the heating rate. The insets of figs. 5(c) and (d) show the plot of $\ln(T_m^2/b)$ vs. $1/T_m$ for the DBCO and YBCO samples. The obtained E_a and τ_0 values for the DBCO and YBCO samples are $E_a = 0.0561 \text{ eV}$; $\tau_0 = 9.19 \times 10^{-7} \text{ s}$ and $E_a = 0.064 \text{ eV}$; $\tau_0 = 7.35 \times 10^{-8} \text{ s}$, respectively. The sign of TSFCC is determined by the type of the trapped charges during the E_p . In the present case, the TSFCC probably originates from the trapped holes, therefore it has the same sign as the E_p , as shown in figs. 5(a) and (b), which is similar to the DyMnO₃ [39].

Discussion and conclusions. – YBCO and DBCO samples belong to the noncentrosymmetric space group similar to the CBCO but the former samples show the absence of spin-driven electric polarization. The main contrast is in the magnetic ground state, which is strongly associated with the strength of the orthorhombic distortion (D). At low temperature, the orthorhombic distortion (D in %) is 0.16, 0.34 and 1.82 for the DBCO, YBCO [19] and CBCO [15] samples, respectively. The high D values in the CBCO leaves the inherent magnetic frustration in the ab plane (or kagome layer) and drives the long-range 3D ferrimagnetic ordering below 70 K. On the contrary the low D values in the DBCO and YBCO samples suggest that the magnetic frustration in the kagome layers remain unchanged down to low temperature and cause the short-range magnetic ordering and glassy behavior in both samples. Apart from giant structural distortions, the other major difference between the CBCO and the YBCO and DBCO might be the cobalt charge ordering in the kagome layers. In CBCO, it is shown that the kagome layer contains the mixed valances of Co^{2+} , Co^{3+} or $\text{Co}^{2+}(L)$ and causes the spin-driven electric polarization [10,40]. On the contrary, according to the nominal composition of the YBCO and DBCO cobaltites, the kagome layers might contain the Co^{2+} valance state, which states the absence of the cobalt charge ordering at kagome layers in both samples.

In conclusion, this study clearly shows a close correlation between the orthorhombic distortion, cobalt charge ordering and spin-driven electric polarization in this RBaCo₄O₇ ($R = \text{Ca}, \text{Y}$ and rare earth) cobaltite family. The magnetization studies on YBaCo₄O₇ show the structural transition ($T_{st} = 300 \text{ K}$), AFM ordering ($T_N = 80 \text{ K}$) and cluster glass behavior ($T_g = 3.6 \text{ K}$). NPD studies on DyBaCo₄O₇ show evidence for the short-range correlations below 70 K and ac magnetization studies show spin glass behavior. Both samples show the absence of spin-induced ferroelectric polarization and extrinsic pyroelectric peaks with the thermally stimulated depolarization currents. This study further indicates that a similar kind of extrinsic electric polarization properties

would be expected from other rare-earth members of this family.

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The authors of IIT Kharagpur acknowledge DST, India for FIST project and IIT Kharagpur funded VSM SQUID magnetometer. We thank RIPANDEEP SINGH, Solid State Physics Division, Bhabha Atomic Research Centre, Mumbai, for his assistance in recording the neutron diffraction data.

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